

SeCl₄ graphite intercalation compound: a precursor material for encapsulated selenium nanoparticles

This article has been downloaded from IOPscience. Please scroll down to see the full text article. 1999 J. Phys.: Condens. Matter 11 L21 (http://iopscience.iop.org/0953-8984/11/5/001)

View the table of contents for this issue, or go to the journal homepage for more

Download details: IP Address: 171.66.16.214 The article was downloaded on 15/05/2010 at 06:56

Please note that terms and conditions apply.

LETTER TO THE EDITOR

SeCl₄ graphite intercalation compound: a precursor material for encapsulated selenium nanoparticles

J Walter[†] and H Shioyama

Osaka National Research Institute, AIST, MITI, 1-8-31 Midorigaoka, Ikeda, Osaka 563-8577, Japan

Received 10 November 1998

Abstract. Selenium tetrachloride (SeCl₄) has been intercalated from the gas phase in the presence of chlorine gas into highly orientated pyrolytic graphite. A mixture of second- and third-stage SeCl₄ graphite intercalation compounds was obtained. X-ray diffraction measurements presented evidence that the *b*-axis of SeCl₄ is orientated perpendicular to the basal plane of the graphite. X-ray diffraction data gave evidence that SeCl₄ forms monolayers inside the host lattice. A selected-area electron diffraction study of the in-plane lattice showed (*h0l*) reflections of SeCl₄. The axis of the guest is commensurate with the *a*-axis of graphite. The lattice parameters for the intercalated species are nearly identical to those of pristine SeCl₄.

Incorporation of nanoparticles or nanorods into a carbon host lattice is of great interest as regards the basic science. A common route for preparing such materials is that of evaporating carbon and metals with an arc discharge in an inert atmosphere [1]. This technique leads mainly to nanometre-sized carbide particles inside the host material [1–3]. The intercalation of alkali metals can be directly performed from the gas phase [4]. A more unusual way is to use metal chloride graphite intercalation compounds (GICs) [5, 6] as precursor materials and to reduce these compounds by means of hydrogen gas [7]. This method has three advantages compared to the more common arc discharge technique:

- (a) Due to the highly reductive atmosphere, it is possible to form metallic nanoparticles and nanorods of tantalum [7]. The conventional synthesis via the arc discharge method produces TaC nanoparticles and nanorods inside the host lattice [3].
- (b) The precursor material has alternating graphene (IUPAC nomenclature for a single carbon layer with graphite structure) and metal chloride layers. Due the formation of monolayers of guests between two sheets of graphene, the reduced compounds contain real twodimensional particles. EXAFS measurements on Ni GICs gave evidence of monolayers of Ni between the graphene sheets [8].
- (c) The host material for intercalation reactions of metal chlorides can be either graphite flakes or carbon fibres [9]. Larger amounts of materials can be synthesized through reduction of metal chloride GICs.

The direct incorporation of selenium into carbon by a CVD process failed [10]. However, a wide variety of metal chlorides (e.g. NbCl₅ [11], TaCl₅ [12], CuCl₂ [13], NiCl₂ [14],

0953-8984/99/050021+05\$19.50 © 1999 IOP Publishing Ltd

[†] Author to whom any correspondence should be addressed. Telephone: +81 727 519615; fax: +81 727 519622; e-mail address: walter@onri.go.jp.

L22 Letter to the Editor

BiCl₃ [15]) can be intercalated into a carbon host. Such metal chloride GICs are classical twodimensional electrical conductors. The lattice parameters of intercalated and pristine metal chlorides are not necessarily identical [16]. Selenium chloride GICs were not described in the literature. They could act as precursors for preparing selenium nanoparticles inside a carbon host lattice. In the present work we characterize a SeCl₄ GIC.

Selenium tetrachloride (SeCl₄) was mixed with highly orientated pyrolytic graphite (HOPG) from Advanced Ceramics Corporation, USA, in a glass ampoule. After evacuation, chlorine gas was condensed in it (under 450 mbar at room temperature) and the ampoule was sealed. The mixture was heated for four days at 190 °C. The success of the intercalation was demonstrated by x-ray diffraction (XRD) and the in-plane superlattice was estimated by selected-area electron diffraction (SAED).



Figure 1. The XRD pattern of SeCl₄ GIC; Cu K $\langle \alpha \rangle$ radiation, 40 kV accelerating voltage, 100 mA anode current. A mixture of a second-stage SeCl₄ GIC ($I_c = 1623 \text{ pm} \pm 55 \text{ pm}$) and a third-stage SeCl₄ GIC ($I_c = 1962 \text{ pm} \pm 18 \text{ pm}$) was obtained; unreacted graphite was not observed. The shape of the reflections, the shifts, the intensity ratios and the few missing reflections indicate a highly disordered sample [18].

Pristine SeCl₄ crystallizes in a monoclinic crystal system with the following lattice parameters [17]: a = 1649 pm, b = 975 pm, c = 1490 pm, $\beta = 117.0^{\circ}$. Figure 1 shows the XRD pattern of a SeCl₄ GIC; the pattern shows a disordered sample, a common result for layered materials such as GICs [18], clays and micas. Disordered GICs can show broad and sharp (00*l*) reflections for one identity period; weaker reflections can be totally absent; some reflections of one identity period can be shifted [18]. A mixture of a second-stage SeCl₄ GIC ($I_c = 1623$ pm \pm 55 pm) and a third-stage SeCl₄ GIC ($I_c = 1962$ pm \pm 18 pm) was formed. Through the knowledge of the lattice parameters of pristine SeCl₄, it can be determined that



Figure 2. A schematic sketch of a graphite host lattice with the axes at the top and the relative orientation of the guest lattice between the sheets of graphene. The *b*-axis of the guest lattice acts as a spacer for the host lattice (c_{gr} is parallel to b_{SeCl_4}).

the guests are orientated with their b-axes perpendicular to the basal plane of the graphite (figure 2). Due to this orientation of the guests, the small axis of the guest crystal acts as a spacer for the host lattice:

$I_{\text{guest}} = I_{c(\text{second-stage GIC})} - I_{c(\text{graphite})} = 1623 \text{ pm} - 670 \text{ pm} = 953 \text{ pm} \approx I_{b(\text{SeCl}_4)}$

The standard deviation of the identity period is large due the high degree of disordering in the sample, ± 55 pm. Taking this standard deviation into consideration, the estimated value of 953 pm ± 55 pm fits very well with the *b*-axis of SeCl₄ (*b* = 975 pm). The interlayer spacing provides evidence that a monolayer of SeCl₄ is incorporated between two graphene sheets.

Figure 3 shows the in-plane diffraction pattern estimated from SAED. From the XRD results it is now evident that this pattern shows (h0l) reflections. The SAED image was obtained at a magnification of 300.000. It is complex, and shows domain twinning of two mirrored lattices (represented by domains A and B). Domain twinning is a common feature and was observed in other GICs by means of SAED [19, 20].

The orientations of the $a_{SeCl_4}^*$ and $c_{SeCl_4}^*$ -axes for both domains are indicated in figure 3. The angle between $a_{SeCl_4}^*$ and $c_{SeCl_4}^*$ is $\beta^* = 180^\circ - \beta = 63^\circ \pm 0.5^\circ$, which is the same as that in the pristine compound. Although the a_{gr}^* - and $c_{SeCl_4}^*$ -axes coincide, due to the different crystal systems (hexagonal: $a_{gr} = h \times d_{(h00)gr}$; monoclinic: $c_{SeCl_4} = (l \times d_{(00l)SeCl_4})/\sin \beta$), the reflections do not superimpose. The two different sets of reflections can be clearly distinguished. The c_{SeCl_4} -axis shows a close relationship with the underlying graphene lattice $(a_{gr} = 245.4 \text{ pm})$: $c_{SeCl_4} = 6 \times a_{gr} = 1533 \text{ pm} \pm 48 \text{ pm}$ (pristine SeCl_4: c = 1490 pm); this axis coincides with the graphite axis. A slight misfit between the host lattice and the a_{SeCl_4} -axis



Figure 3. The SAED pattern of the in-plane lattice; 300 kV accelerating voltage, 1 m camera length. The pattern can be interpreted as a superposition of two reciprocal lattices, due to the twinning of domain A and domain B. The SAED image shows $(h0l)_{SeCl_4}$ reflections; the $a_{SeCl_4}^*$ and $c_{SeCl_4}^*$ -axes of the guest lattice are indicated, as well as the axis of the graphite lattice (a_{gr}^*) . The reciprocal angle between the two axes of the SeCl_4 lattice is $\beta^* = 180^\circ - \beta = 63^\circ \pm 0.5^\circ$, identical to that in pristine SeCl_4. A mirror plane (m) exists between the two lattices; the centre acts as an inversion point (i). The following symbols are used: * indicates $(h00)_{gr}$ reflections; # indicates $(00l)_{SeCl_4}$ reflections; + indicates $(h00)_{SeCl_4}$ reflections.

was detected, as regards their lattice constants: $a_{\text{SeCl}_4} = 1633 \text{ pm} \pm 32 \text{ pm} \approx 7 \times a_{\text{gr}}$ (pristine SeCl₄: 1652 pm), and as regards their slightly different angles ($\beta_{\text{SeCl}_4}^* = 63^\circ \pm 0.5^\circ$, $\beta_{\text{gr}}^* = 60^\circ$).

Selenium tetrachloride was successful intercalated into graphite. The lattice parameters of the guest particles show close relationships with the lattice parameters of the pristine metal chloride. Monolayers of SeCl₄ are incorporated between sheets of graphene. Because these monolayers are present between the graphene sheets, SeCl₄ GIC can act as a precursor material for preparation of two-dimensional selenium nanoparticles inside the carbon host lattice. Nanoparticles in the quantum-size regime can show uncommon magnetization due a quantum-size effect [21]. Two-dimensional selenium nanoparticles will be prepared from this SeCl₄ GIC precursor.

Jürgen Walter is grateful to the Alexander von Humboldt-Foundation (AvH, Germany), and to the Science and Technology Agency (STA, Japan) for his fellowship in Japan.

References

- [1] Ruoff R S, Lorents D C, Chan B, Malhotra R and Subramoney S 1993 Science 259 346
- [2] Murakami Y, Shibata T, Okuyama K, Arai, T, Suematsu H and Yoshida Y 1993 J. Phys. Chem. Solids 54 1861
- [3] Fukunaga A, Chu S and McHenry M E 1998 J. Mater. Res. 13 2465
- [4] López S, Granados S and Mendoza D 1996 J. Phys.: Condens. Matter 8 L435
- [5] Walter J and Maetz M 1997 Mikrochim. Acta 127 183
- [6] Walter J 1997 Synth. Met. 89 39
- [7] Walter J, Shioyama H and Sawada Y 1999 Carbon 37 41

- [8] Shuvage A T, Helmer B Y, Lyubenova T A, Kraizman V L, Mirmilstein A S, Kvacheva L D, Novikov Y U and Volpin M E 1989 J. Physique 50 1145
- [9] Walter J and Shioyama H 1998 Synth. Met. 92 91
- [10] Mendoza D, Lopez S, Granados S, Morales F and Escudero R 1997 Synth. Met. 89 71
- [11] Walter J and Boehm H P 1995 Carbon 33 1121
- [12] Walter J 1997 Solid State Ion. 101-103 833
- [13] Capkova P, Rafaja D, Walter J, Boehm H P, van Mallsen K F and Schenk H 1995 Carbon 33 1425
- [14] Stumpp E and Werner F 1966 Carbon 4 538
- [15] Walter J, Shioyama H and Sawada Y 1998 Carbon 36 1811
- [16] Behrens P, Woebs V, Jopp K and Metz W 1988 Carbon 26 641
- [17] Golubkova G, Petrov E and Samsonova T 1976 Zh. Strukt. Khim. 17 276
- [18] Hohlwein D and Metz W 1974 Z. Kristallogr. 139 279
- [19] Walter J, Shioyama H, Sawada Y and Hara S 1998 Carbon 36 1277
- [20] Ghosh D and Chung D D L 1983 Mater. Res. Bull. 18 727
- [21] Volokitin Y, Sinzig J, de Jongh L J, Schmid G, Vargaftik M N and Moiseev I I 1996 Nature 384 621